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Fabrication and characterization of three-dimensional all metallic photonic crystals for near infrared applications

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Three-dimensional all metallic photonic crystals with a feature size of 0.20 μm were fabricated using electron-beam lithography with the photoresist of hydrogen silesquioxane. This process method has high compatibility with the fabrication of damascene copper interconnections and also simplifies the whole process flow. The dependence of the complete photonic band gaps on polarization was experimentally observed and compared with the simulation results. The band edge for the four-layer lattice was found at a wavelength of around 0.80 μm in normal incidence. 

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In recent years, three-dimensional (3D) photonic crystals (PCs) have attracted great attention since they have the ability to control the flow of light omnidirectionally for a specific range of frequencies, which is so-called a complete photonic band gap (PBG). Among a variety of the proposed 3D structures, the woodpile structure1 has been explored for its relatively large PBG and simple design rules of introducing the intentional defects.2–4 This structure has been successfully realized with dielectric materials by some state-of-the-art technologies, such as wafer bonding,5 direct laser writing,6,7 and atomic layer deposition.8 For their metallic counterpart, most demonstrations are limited to the 3D metallic PCs created by performing the different metallization techniques on the 3D dielectric templates.9–12 However, fabricating a 3D all metallic PC with a feature size of submicron in a large area is still a challenge due to its complex processing steps.

One of the promising features for metallic PCs is the capability of tailoring their thermal emission spectra. In the range of PBG, thermal emission from the 3D metallic PCs is suppressed, and narrow band emission is observed at the wavelengths near the photonic band edges.13,14 These metallic PCs have been demonstrated as light sources with high electrical-to-optical conversion efficiency. To achieve emission at shorter wavelengths, the feature sizes of the 3D lattices are necessary to be scaled down for shifting the band edges to higher frequencies. In addition, choosing appropriate materials to construct metallic PCs plays an important role as well. For some metals available in standard integrated circuit (IC) technology, severe intrinsic absorption at near infrared or visible wavelengths is an obstacle to scale the band edge down. It is essential to adopt a metal of high plasma frequency for holding the linear scaling rule of the band edge positions.9

In this letter, copper was selected as the material to form the 3D lattice with a feature size of 0.20 μm for its desirable optical properties at near infrared wavelengths and high compatibility with the fabrication of damascene copper interconnections.9,15 To simplify the complexity of processes, the electron-beam (e-beam) lithography was used to realize the 3D all copper PCs with the photoresist of hydrogen silesquioxane (HSQ). Experimentally, the complete photonic band edge was observed at a wavelength of around 0.80 μm. Its dependence on polarization of incident light will be discussed.

The 3D PCs that we realized are the four-layer woodpile structure, similar to face-centered-tetragonal (fct) lattice symmetry. For the ideal fct lattice, they consist of parallel copper rods with a pitch of a in each layer, where a =0.50 μm in this report. The orientation of rods in one layer is perpendicular to that in the adjacent layers. The position of the rods in one layer is shifted by a/2 with respect to that in the alternate layers. As shown in Fig. 1, practically, the copper rods in the third and fourth layers are shifted by

![FIG. 1. (a) A SEM image of the HSQ pattern formed after the process of e-beam lithography. (b)–(d) SEM images of the 3D all copper PCs from the tilted, top, and cross-sectional views, respectively. The width and height of copper rods are about 200 and 180 nm, respectively. The rod-to-rod spacing is about 500 nm. This structure was built with four layers to form one unit cell of a fct-like lattice. The deviation from the ideal fct lattice is due to the misalignment of copper rods in the third and fourth layers.](image)
0.05 \mu m with respect to the ideal position. The 3D copper PCs were fabricated layer-by-layer using the process of e-beam lithography. First, the HSQ film (FOX-15, Dow Corning) was spin-coated on a silicon wafer as a high-resolution negative photoresist for the e-beam lithography (Weprint 200 type, Leica). After the processes of exposure and development, the grating pattern was formed as shown in Fig. 1(a). The HSQ pattern was annealed at 400 \degree C for 1 h in order to resist the following process of chemical-mechanical polishing (CMP). The 25 nm thick tantalum and 50 nm thick copper seed layers were then deposited on the resulting HSQ pattern by the physical vapor deposition system. This step was followed by the processes of copper electroplating for filling the HSQ trenches and CMP for removing excess copper and tantalum on the top of the HSQ trenches. The same processes were repeated layer-by-layer as desired. After this HSQ mold was removed by a solution of dilute hydrofluoric acid, the entire fabrication processes were completed. This etchless method is compatible with the copper interconnect technology and simplifies the whole process flow owing to omission of processes of extra depositing and etching intermetal dielectric layers, which allows easier formation of patterns with smaller feature sizes (\approx 100 \, nm). The HSQ layers were used as the photoresist of the e-beam writer as well as the sacrificial dielectric mold for the metallic PCs. The scanning electron microscope (SEM) images of the fabricated 3D all copper PCs are shown in Figs. 1(b)–1(d). The width and height of the fabricated copper rods are about 0.20 and 0.18 \mu m, respectively. The overall sample area is 5 \times 5 \, \text{mm}^2.

To observe the PBG behaviors of the fabricated 3D all copper PCs, the Fourier transform infrared (FTIR) microscope (Hyperion 2000, Bruker) was used to measure the reflectance spectra for the two-, three-, and four-layer structures. The reflectance was obtained by normalizing the measured results taken from the fabricated samples to that from the uniform gold mirror. In Fig. 2, the measured reflectance spectra in the case of unpolarized incident light are shown and compared with the simulation results for s- and p-polarization, which were taken for an ideal four-layer fct lattice with the same dimensions as the practical one. This simulation was performed by the 3D finite-difference time-domain method with the commercial software (Rsoft), and the wavelength-dependent refractive indices were used. As shown in Fig. 2, the PBG develops gradually as the number of layers increases. For the four-layer lattice, the measured reflectance keeps very high values (\approx 0.95) for wavelengths longer than 2.95 \mu m due to the typical PBG effect of the metallic PCs, exhibits a dip at a wavelength of about 1.20 \mu m, and decreases sharply to 0.31 at a wavelength of 0.60 \mu m. The sharp drop for the wavelengths shorter than 0.60 \mu m is caused by the increase in the absorption of copper and the transmission for the waveguide effect with the cutoff wavelength of 0.60 \mu m. The photonic band edge, where reflectance is defined as 0.5, was found at a wavelength of about 0.80 \mu m. The measured results show the similar trend to the simulation results except for the lower reflectance values at wavelengths near the band edge. The dips in the simulation results at a wavelength of around 0.80 \mu m correspond to the narrow transmission and absorption peaks. In the measurements, these dips are not clearly observed, which might be due to the deviations between the ideal lattice and the actual samples. Possible causes of the dip at the wavelength of about 1.20 \mu m are as follows. First, copper is of intrinsic absorption in the wavelength ranges from about 1.30 to 1.70 \mu m, as shown in the inset of Fig. 2. The reflectance R of the copper and tantalum films was calculated as \[ R = \frac{(n-1)^2 + k^2}{(n+1)^2 + k^2}, \] where n and k are the real and imaginary parts of the wavelength-dependent refractive index of metal, respectively. The absorptance A was assumed to be 1−R (for no transmission). Second, owing to the limited acceptance angle of the objective lens equipped in the FTIR microscope, only the portion of the Rayleigh scattered light from the 3D lattices captured by the objective lens contributes to the measured signals, leading to the discrepancy between the measured and simulation results. It is inferred that this loss of magnitude becomes more significant as the number of PC layers increases and wavelengths of measured incident light are shorter. In comparison with the copper-coated counterpart reported by Tal et al., our structure has advantages in higher compatibility with the standard IC technology and capability for designing more functionalities in one chip (larger lateral areas) and with a smaller feature size for achieving the photonic band edge to a shorter wavelength.

To obtain insight into the characterization of the 3D copper PCs, the reflectance spectra for s- and p-polarized incident light were measured and shown in Fig. 3. The reflectance spectra are depicted as a function of number of layers. Figure 3(a) shows that measured reflectances for different layer structures are roughly identical for s-polarization. This is because the waveguide cutoff effect plays an important role for this case. The slight drop appearing at a wavelength of about 2.29 \mu m for the four-layer lattice, which was not predicted from the simulation results, may stem from the imperfections in the sample. On the other hand, for p-polarization, the band edges are shifted to the shorter wavelengths when the number of layers increases, as shown in Fig. 3(b). This is the evidence for the development of the complete band gap for the woodpile PCs. It is noted that the dip at a wavelength of about 1.20 \mu m in the measured spectrum for the four-layer lattice is much deeper than that from the simulation results. This may be attributed to the fact that...
for the $p$-polarized incident light, the longer penetration length into the 3D copper PCs leads to stronger Rayleigh scattering at large angles and lower measured magnitude detected by the FTIR microscope. Consequently, the measured reflectance spectra represent the combined results of the loss of reflected signals from samples and the PBG effect of the 3D metallic PCs.

To verify the properties of the complete PBG, the tilted-angle measurements were performed with the grazing angle objective (Bruker) equipped in the FTIR microscope, as shown in Fig. 4. The angles of grazing incidence are between 52.2° and 84.2° off the normal to the surface of the sample, as shown in the inset of Fig. 4. The measured results show that the reflectance still keeps high values (≥0.80) at wavelengths longer than 3.25 μm for both polarizations and decreases sharply at shorter wavelengths. The locations of band edges are shifted to the longer wavelengths owing to the shortage of the number of layers. In addition, the tantalum layers adhered on the side walls of copper rods in each layer may play an important role for the results of grazing-angle measurements. Tantalum is absorptive at wavelengths shorter than 1.50 μm as shown in the inset of Fig. 2. The incident light at such large angles experiences more tantalum, which leads to the more significant drop in reflectance. Therefore, for the four-layer structure, the edge of the complete PBG is at a wavelength of around 1.50 μm in grazing-angle incidence, implying that the 3D copper PCs can be used for the near infrared applications.

In summary, the e-beam lithography with the photoresist of the HSQ has been adopted to create the 3D all copper PCs with a feature size of 0.20 μm. This approach is compatible with the standard IC technology and also simplifies the whole process flow. Experimental results show that the photonic band edge is located at a wavelength of around 0.80 μm in normal incidence. For applications in the visible wavelength range, we offer a possible method to accomplish all metallic PCs with smaller feature sizes.

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